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# Intrinsic catalytic activity of SiO<sub>2</sub>-supported Co–Mo and Co–W sulfide catalysts for the hydrodesulfurization of thiophene

Yasuaki Okamoto\*, Akira Kato, Usman, Kenji Sato, Ichiro Hiromitsu, Takeshi Kubota

Department of Material Science, Shimane University, Matsue 690-8504, Japan Received 13 January 2005; revised 28 March 2005; accepted 1 April 2005 Available online 23 May 2005

#### Abstract

SiO<sub>2</sub>-supported Co–Mo and Co–W sulfide catalysts with varying Mo(W) content were prepared from Mo(W)S<sub>2</sub>/SiO<sub>2</sub> presulfided at 673–873 K with a CVD technique using Co(CO)<sub>3</sub>NO as a precursor to Co sulfide. The Al<sub>2</sub>O<sub>3</sub>-supported counterparts were also briefly studied. The catalysts were characterized by NO adsorption, TEM, XPS, W L<sub>3</sub>-edge XANES, and the magnetic measurements of Co. The Co atoms in the Co–W–S phase showed an antiferromagnetic property. The TOF of the hydrodesulfurization of thiophene on the Co–Mo(W)–S phase significantly increased as the presulfidation temperature of Mo(W)S<sub>2</sub>/SiO<sub>2</sub> increased from 673 to 873 K, whereas it was independent of the loading of Mo(W). The TOFs on the Co–Mo–S phase and the Co–W–S phase were close to each other but strongly dependent on the support. It is suggested that complete sulfidation and the formation of stacked Mo(W)S<sub>2</sub> particles do not necessarily result in the formation of Co–Mo(W)–S Type II.

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#### 1. Introduction

Development of highly active hydrodesulfurization (HDS) catalysts has been one of the most urgent issues in the petroleum industry related to protection of the environment [1–3]. Supported Mo sulfides promoted by Co or Ni have been extensively used in industry for the HDS reaction [4–7]. Topsøe and co-workers [5,8–11] have proposed that the catalytically active phase in the promoted catalysts is the so-called Co(Ni)–Mo–S phase (structure), in which Co(Ni) atoms are located on the edge of MoS<sub>2</sub> particles. The molecular structure of the Co(Ni)–Mo–S phase and the reaction mechanism thereon are still a matter of debate [12–22].

Supported W sulfide catalysts promoted by Ni have also been widely studied because of the generation of strong catalytic synergies between W and Ni [4–7]. On the other hand, Co–W sulfide catalysts have received much less attention compared with Co–Mo catalysts because of weaker promotional effects of Co in conventionally prepared Co–W catalysts [23]. Recent studies [24,25], however, have shown that this is due, at least in part, to a greater difference in the sulfidation temperature between Co and W than that between Co and Mo, resulting in a lower coverage of Co on the edge of WS<sub>2</sub> particles than that on the edge of MoS<sub>2</sub> particles. The addition of a chelating agent to the impregnation solution of Co increased the sulfidation temperature of Co by about 150 K because of Co-chelate complex formation, accompanying a significant increase in the HDS activity of Co–W catalysts [25]. Nevertheless, the reason for the weak catalytic synergy between Co and W sulfides is not clear at present.

The HDS activity of Co-promoted Mo or W catalysts is determined by the amount of the Co–Mo(W)–S phase and its intrinsic activity. It is therefore intriguing to compare the intrinsic activities of the Co–Mo–S phase and the Co–W–S phase for HDS reaction in order to understand the nature of Co–W catalysts and to obtain important insights into the role

<sup>\*</sup> Corresponding author. Fax: +81 852 32 6466.

E-mail address: yokamoto@riko.shimane-u.ac.jp (Y. Okamoto).

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of Co situated on the  $MoS_2$  or  $WS_2$  edge in the HDS reaction thereon, and into the role of the  $MoS_2$  or  $WS_2$  edge. With carbon-supported Ni–Mo and Ni–W sulfide catalysts, Louwers and Prins [26] suggested that the intrinsic catalytic activities of the Ni–Mo–S phase and the Ni–W–S phase are identical for the HDS of thiophene. It seems, however, that there remain some ambiguities about their assumptions that all of the Ni atoms are situated on the edge of  $MoS_2$  or  $WS_2$ particles in the catalyst systems, since no sufficient characterizations were presented, except for EXAFS.

For the purpose of comparing the intrinsic catalytic activities of the Co(Ni)-Mo-S phase and the Co(Ni)-W-S phase, it is very desirable to prepare Co(Ni)-Mo and Co(Ni)-W catalysts, in which the edges of MoS<sub>2</sub> or WS<sub>2</sub> particles are selectively covered by the Co(Ni)-Mo-S phase or the Co(Ni)-W-S phase without a concomitant formation of Co(Ni) sulfide clusters with a low catalytic activity. In a previous study [27-30] we showed that supported Co-Mo sulfide catalysts, in which the edge of MoS<sub>2</sub> particles is fully covered by Co atoms forming the Co-Mo-S phase, can be selectively prepared when sulfided Mo catalysts are exposed to Co(CO)<sub>3</sub>NO vapor (CVD technique), followed by evacuation and resulfidation. The Co-Mo catalysts thus prepared by the CVD technique are designated CVD-Co/MoS<sub>2</sub>/support here. The amount of Co accommodated in CVD-Co/MoS<sub>2</sub>/support (support: Al<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub>,  $ZrO_2$ , or  $SiO_2$ ) is proportional to the amount of NO adsorption on the MoS<sub>2</sub>/support, suggesting that Co atoms are preferentially located at the edge of MoS<sub>2</sub> particles [27]. The Co2p XPS [27] and Co K-edge XANES [30] spectra clearly demonstrated a selective formation of Co-MoS<sub>2</sub> interaction species. The FTIR results of CO adsorption by Maugé et al. [31] also suggested the selective formation of the Co-Mo-S phase in ex-carbonyl catalysts. However, no clear experimental evidence was reported about selective formation of the Co-W-S phase by the CVD technique, except for spectral indications from Co K-edge XANES [25].

In the present study, we prepared CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> with a varying Mo or W loading to compare the intrinsic activities of the Co–Mo–S phase and the Co–W–S phase for the HDS of thiophene. The effect of presulfidation temperature was also examined. We studied mainly SiO<sub>2</sub>-supported Co–Mo(W) sulfide catalysts and briefly Al<sub>2</sub>O<sub>3</sub>-supported counterparts, even though the latter catalysts have been widely employed as practical HDS catalysts, to avoid strong and complicated interactions between the active phase and support and thereby to directly extract the nature of the active sites. It is concluded that the turnover frequencies (TOFs) of the HDS of thiophene on the Co–Mo–S phase and the Co–W–S phase are close to each other and significantly depend on the presulfidation temperature and support.

#### 2. Experimental

#### 2.1. Catalyst preparation

A series of SiO<sub>2</sub>-supported MoO<sub>3</sub> catalysts (MoO<sub>3</sub>/SiO<sub>2</sub>) with 5, 10, 15, and 20 wt% MoO<sub>3</sub> was prepared by an impregnation technique using (NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub> · 4H<sub>2</sub>O (Kanto Chemicals; analytical grade). After evaporation of the impregnation solution to dryness at ca. 350 K with stirring, the catalyst was dried at 373 K for 16 h. The catalyst was calcined in air at 773 K for 5 h with an electric furnace. A corresponding series of WO<sub>3</sub>/SiO<sub>2</sub> catalysts (5, 10, 15, and 20 wt% WO<sub>3</sub>) was prepared in an analogous way with (NH<sub>4</sub>)<sub>6</sub>W<sub>12</sub>O<sub>39</sub> · 5H<sub>2</sub>O (Kanto Chemicals; analytical grade) as a starting precursor. The SiO<sub>2</sub> supports (JRC-SIO-8, 323 m<sup>2</sup> g<sup>-1</sup> for MoO<sub>3</sub>/SiO<sub>2</sub>, and JRC-SIO-9, 370 m<sup>2</sup> g<sup>-1</sup> for WO<sub>3</sub>/SiO<sub>2</sub>) were supplied by the Catalysis Society of Japan as reference catalysts.

The MoO<sub>3</sub>/SiO<sub>2</sub> catalyst thus prepared was presulfided at 673, 773, or 873 K in a 10%  $H_2S/H_2$  flow at an atmospheric pressure for 1.5 h (673 K) or 2 h (773 and 873 K). The Mo sulfide catalyst is designated MoS<sub>2</sub>/SiO<sub>2</sub>, followed by the presulfidation temperature in parentheses if necessary. WO<sub>3</sub>/SiO<sub>2</sub> was sulfided at 673, 773, or 873 K for 2 h to prepare WS<sub>2</sub>/SiO<sub>2</sub>.

WO<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> (22 wt% WO<sub>3</sub>) and MoO<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> (13 wt% MoO<sub>3</sub>) were prepared in a similar way. The surface area of the Al<sub>2</sub>O<sub>3</sub> support (JRC-ALO-7) was 180 m<sup>2</sup> g<sup>-1</sup>. MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> and WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> were presulfided at 673 or 773 K for 1.5 and 2 h, respectively.

Cobalt was introduced to MoS<sub>2</sub>/SiO<sub>2</sub> and WS<sub>2</sub>/SiO<sub>2</sub> by a CVD technique [27–30]. Briefly, MoS<sub>2</sub>/SiO<sub>2</sub> was first evacuated at 673 K for 1 h and subsequently exposed for 5 min at room temperature to a vapor of Co(CO)<sub>3</sub>NO (Strem Chemicals) kept at 273 K (CVD technique). After evacuation for 10 min at room temperature, the sample was sulfided again at 673 K for 1.5 h to prepare CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> regardless of the presulfidation temperature of MoS<sub>2</sub>/SiO<sub>2</sub>. CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub>, CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub>, and CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> were prepared in a similar manner. CVD-Co/SiO<sub>2</sub> was prepared similarly with the use of presulfided SiO<sub>2</sub> (JRC-SIO-9). The amount of Co incorporated by the CVD technique was determined by means of XRF (Rigaku RIX2000) within an accuracy of 5% [30].

#### 2.2. Reaction procedure

The initial activity of the freshly prepared catalyst for the HDS of thiophene was evaluated at 623 K with a circulation reaction system made of glass under mild reaction conditions (initial H<sub>2</sub> pressure, 20 kPa). The HDS activity was calculated on the basis of the accumulated amount of H<sub>2</sub>S. The details of the reaction procedures have been reported previously [27].

#### 2.3. Characterization

The amount of NO adsorption on the catalyst was measured at room temperature by a pulse technique after the catalyst sample was cooled from the presulfidation temperature in a 10% H<sub>2</sub>S/H<sub>2</sub> stream. The sample was purged for 15 min with a He stream before periodic admissions of a pulse of 10% NO/He. Details of the procedures have been reported elsewhere [19,27].

The W  $L_3$ -edge X-ray absorption near-edge structure (XANES) spectra of the catalysts and reference compounds were measured in transmission mode at room temperature at BL-10B of KEK-IMSS-PF with the use of an in situ XAFS cell with Kapton windows, with a 2.5-GeV ring energy and 250–290 mA of stored current.

The static magnetic susceptibility measurement of the sulfided catalyst was made in situ with a Faraday method with the use of a Cahn 2000 Electro-Balance system between 4.2 and 300 K [19]. The catalyst sample was evacuated at 673 K for 1 h (ca.  $10^{-2}$  Pa) before it was fused into a glass ampoule (Al<sub>2</sub>O<sub>3</sub>-supported catalysts) or a quartz ampoule (SiO<sub>2</sub>-supported catalysts). The magnitude of the magnetic field was fixed at 10,000 G. We obtained the effective magnetic moment and magnetic susceptibility by subtracting the magnetic contributions of the glass or quartz ampoule and the corresponding supported MoS<sub>2</sub> or WS<sub>2</sub> catalyst or Al<sub>2</sub>O<sub>3</sub> measured separately under identical conditions.

The XPS spectra of 20% WS<sub>2</sub>/SiO<sub>2</sub> (673) were measured on an X-ray photoelectron spectrometer, an ESCA 5400 (ULVAC-PHI). The catalyst was mounted on double-sided adhesive tape in an N<sub>2</sub>-filled glove box and transferred to the pretreatment chamber with the use of a specially designed cell to avoid air contamination. Mg- $K_{\alpha_{1,2}}$  radiation (1253.6 eV) was used, and the pass energy was adjusted to 35.75 eV. The binding energies were referenced to the Si2*p* level at 103.8 eV due to SiO<sub>2</sub>.

TEM observations were made on a JEM-2010 (200 kV) for 10%  $MoS_2/SiO_2$  (673) and 20%  $WS_2/SiO_2$  (673). The catalyst sample was dispersed in heptane in a N<sub>2</sub>-filled glove bag. The distributions of  $MoS_2$  or  $WS_2$  slab size and stacking number were calculated over 100–250 particles [28].

#### 3. Results

The catalytic activities of CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> catalysts for the HDS of thiophene are shown in Fig. 1 as a function of Mo or W loading,  $S_{Mo}$  or  $S_W$  (atoms nm<sup>-2</sup>). CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> exhibited a considerably higher HDS activity than CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> over the whole range of metal loading. Regardless of the presulfidation temperature, the HDS activities of CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> weakly depended on  $S_{Mo}$  and  $S_W$ , with broad maximum activities around 2.3 Mo atoms nm<sup>-2</sup> and 0.8 W atoms nm<sup>-2</sup>, respectively.

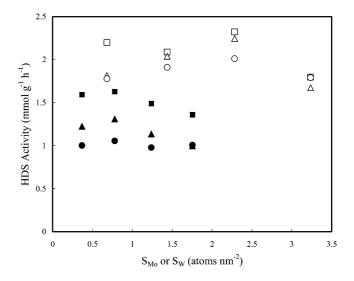


Fig. 1. Catalytic activity of CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (open symbols) and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (closed symbols) for the HDS of thiophene at 623 K as a function of loading (atoms nm<sup>-2</sup>) of Mo ( $S_{Mo}$ ) or W ( $S_W$ ). Presulfidation temperature:  $\bigcirc$  and  $\spadesuit$ , 673 K;  $\triangle$  and  $\bigstar$ , 773 K; and  $\square$  and  $\blacksquare$ , 873 K.

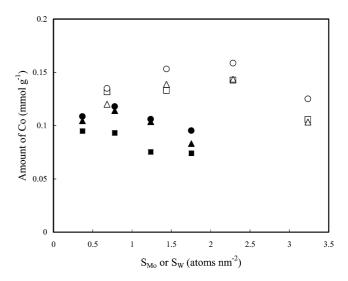


Fig. 2. Amount of Co in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (open symbols) and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (closed symbols) as a function of loading (atoms nm<sup>-2</sup>) of Mo ( $S_{Mo}$ ) or W ( $S_{W}$ ). Presulfidation temperature:  $\bigcirc$  and  $\blacklozenge$ , 673 K;  $\triangle$  and  $\blacklozenge$ , 773 K; and  $\Box$  and  $\blacksquare$ , 873 K.

The HDS activity of CVD-Co/Mo(W)S<sub>2</sub>/SiO<sub>2</sub> increased with increasing presulfidation temperature, particularly at a lower  $S_{Mo}$  or  $S_W$ . CVD-Co/SiO<sub>2</sub> showed only a negligible HDS activity (0.02 mmol g<sup>-1</sup> h<sup>-1</sup>). This is due to a very limited anchoring capacity of Co on SiO<sub>2</sub> by the CVD technique and to the formation of less active Co sulfide clusters [27].

Fig. 2 shows the amount of Co anchored by the CVD technique as a function of  $S_{Mo}$  or  $S_W$ . The amount of Co exhibited a broad maximum around 2.3 Mo and 0.8 W atoms nm<sup>-2</sup> for CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub>, respectively, except for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (873), in which the amount of Co decreased monotonically with increasing W content. The amount of Co in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> decreased slightly as the presulfidation temperature of

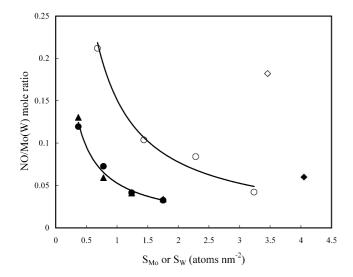


Fig. 3. The NO/Mo(W) mole ratios for MoS<sub>2</sub>/SiO<sub>2</sub> ( $\bigcirc$ ), WS<sub>2</sub>/SiO<sub>2</sub> ( $\blacklozenge$  and  $\blacktriangle$ ), 13% MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> ( $\diamondsuit$ ), and 22% WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> ( $\blacklozenge$ ) as a function of loading (atoms nm<sup>-2</sup>) of Mo ( $S_{MO}$ ) or W ( $S_{W}$ ). Presulfidation temperature:  $\bigcirc$ ,  $\diamondsuit$ ,  $\diamondsuit$ , 673 K and  $\blacktriangle$ , 773 K.

 $MoS_2/SiO_2$  increased from 673 to 773 K, but no further decrease was observed for the catalyst presulfided at 873 K. In CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub>, the amount of Co was decreased considerably by the presulfidation between 773 and 873 K.

For MoS<sub>2</sub>/SiO<sub>2</sub> and WS<sub>2</sub>/SiO<sub>2</sub> catalysts sulfided at 673 or 773 K, the NO/Mo and NO/W mole ratios are plotted in Fig. 3 against  $S_{Mo}$  and  $S_W$ . The NO/Mo and NO/W ratios decreased as  $S_{Mo}$  and  $S_W$  increased, indicating decreasing dispersion of MoS<sub>2</sub> or WS<sub>2</sub> particles. Fig. 3 also presents the NO/Mo and NO/W mole ratios for 13% MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673) and 22% WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673). Apparently, the dispersion of Mo and W sulfides on the Al<sub>2</sub>O<sub>3</sub>-supported catalysts is much higher than that on the SiO<sub>2</sub>-supported counterparts.

The morphology of  $MoS_2$  and  $WS_2$  particles in 10%  $MoS_2/SiO_2$  (673) and 20%  $WS_2/SiO_2$  (673) was observed by TEM (not shown for brevity). The average stacking numbers were calculated as 2.1 and 3.7, and the average slab lengths were 2.7 and 6.7 nm for the Mo and W catalysts, respectively. Obviously,  $MoS_2$  particles are more highly dispersed than  $WS_2$  particles at a similar loading of Mo (1.44 atoms nm<sup>-2</sup>) and W (1.75). In both catalyst systems, stacked  $MoS_2$  and  $WS_2$  particles were predominant.

Fig. 4 shows a correlation between the Co/W mole ratio for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> and the NO/W mole ratio for WS<sub>2</sub>/SiO<sub>2</sub> presulfided at 673 or 773 K. The results for 22% WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> are also shown in Fig. 4. A single proportional line was obtained for the series of WS<sub>2</sub>/SiO<sub>2</sub> and 22% WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> within experimental accuracy, regardless of the presulfidation temperature. A similar proportional correlation was observed for MoS<sub>2</sub>/SiO<sub>2</sub> (673), as shown previously for supported MoS<sub>2</sub> catalysts [27,28]. The correlation in Fig. 4 suggests [27–29] that the Co–W–S phase is selectively formed by the CVD technique, with Co(CO)<sub>3</sub>NO as a precursor to Co sulfides.

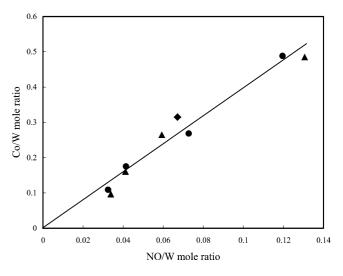


Fig. 4. Correlation between the Co/W mole ratio for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> and the NO/W mole ratio for WS<sub>2</sub>/SiO<sub>2</sub>. The results for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> ( $\blacklozenge$ ) are also included. Presulfidation temperature:  $\bullet$  and  $\blacklozenge$ , 673 K and  $\blacktriangle$ , 773 K.

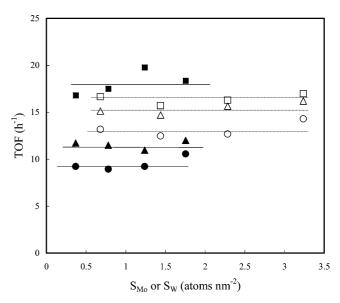


Fig. 5. TOF of the thiophene HDS at 623 K over CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (open symbols) and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (closed symbols) as a function of  $S_{Mo}$  or  $S_W$ . Presulfidation temperature:  $\bigcirc$  and  $\spadesuit$ , 673 K;  $\triangle$  and  $\blacktriangle$ , 773 K; and  $\square$  and  $\blacksquare$ , 873 K.

The intrinsic activity of the catalyst was evaluated in terms of a TOF calculated on the basis of the amount of Co anchored by the CVD technique, assuming that the Co–Mo(W)–S phase is selectively prepared by the present technique. As shown in our previous study [27,29], the HDS reaction on unpromoted Mo(W)S<sub>2</sub> edges in the CVD-Co/MoS<sub>2</sub> catalysts can be neglected. Fig. 5 shows the TOF values over CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> as a function of  $S_{Mo}$  or  $S_W$ . The TOF was almost independent of the surface concentration of Mo or W, except for a slight tendency toward a higher TOF at the highest loading for the Mo and W catalysts presulfided at 673 K. However, the TOF strongly depended on the catalyst component and

Table 1	
TOF of thiophene HDS on the Co-Mo-S phase and the Co-W-S phase supported on SiO <sub>2</sub> and Al <sub>2</sub> O <sub>3</sub>	

Support	Mo/W	Amount of oxides (wt%)	Presulfidation temperature (K)	TOF (h <sup>-1</sup> )	Relative TOF		
					$SiO_2 (T_1)/Al_2O_3(673)^a$	$TOF(T_2)/TOF(673)^b$	
SiO <sub>2</sub>	Mo	5-20	673	12.8	1.8	1.0	
			773	15.4	2.1	1.2	
			873	16.5	2.3	1.3	
SiO <sub>2</sub>	W	5-20	673	9.1	1.3	1.0	
			773	11.2	1.6	1.2	
			873	17.9	2.5	2.0	
$Al_2O_3$	Mo	13	673	7.3	_	1.0	
			773	8.7	_	1.2	
	W	22	673	7.1	_	1.0	
			773	7.7	_	1.1	

<sup>a</sup> Relative TOF is defined as the ratio of the TOF value on the SiO<sub>2</sub>-supported CVD-Co/Mo(W)S<sub>2</sub> catalyst to the one on the corresponding Al<sub>2</sub>O<sub>3</sub>-supported CVD-Co/Mo(W)S<sub>2</sub> catalyst sulfided at 673 K. Presulfidation temperature  $T_1 = 673$ , 773, or 873 K.

<sup>b</sup> Relative TOF is defined as the ratio of the TOF on the CVD-Co/Mo(W)S<sub>2</sub> catalyst sulfided at 773 K to the one in the corresponding CVD-Co/Mo(W)S<sub>2</sub> catalyst sulfided at 673 K. Presulfidation temperature  $T_2 = 773$  or 873 K.

the presulfidation temperature of  $MoS_2/SiO_2$  and  $WS_2/SiO_2$ . Table 1 summarizes the TOF values over  $SiO_2$ - and  $Al_2O_3$ supported CVD-Co/MoS<sub>2</sub> and CVD-Co/WS<sub>2</sub> catalysts presulfided at 673, 773, or 873 K.

The TOF values over CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> and CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> presulfided at 673 K were the same within experimental accuracy, suggesting that the Co-Mo-S phase and the Co-W-S phase exhibit the same intrinsic activity. The high-temperature presulfidation of the Al<sub>2</sub>O<sub>3</sub>-supported CVD-Co/MoS<sub>2</sub> and CVD-Co/WS<sub>2</sub> catalysts caused a 10–20% increase in the TOF. The TOF on CVD-Co/MoS<sub>2</sub>/  $SiO_2$  (673) was 1.8 times as high as that on CVD-Co/MoS<sub>2</sub>/  $Al_2O_3$  (673), in excellent agreement with the previous observations [27]. On the other hand, the TOF on CVD- $Co/WS_2/SiO_2$  (673) was only 1.3 times as high as that on the Al<sub>2</sub>O<sub>3</sub>-supported counterpart. With regard to the SiO<sub>2</sub>supported catalysts presulfided at 673 K, the Co-Mo-S phase showed a considerably higher intrinsic activity than the Co-W-S phase. When the catalysts were presulfided at 773 K, the TOF values on both CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> increased by about 20%. When CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> was presulfided at 873 K, the TOF increased further, but to a smaller extent (7%). However, the effect of the high-temperature presulfidation was very significant for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub>, and the TOF increased by 60% compared with that for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (773). The TOF on the Co-W-S phase was comparable to or even higher than that on the Co-Mo-S phase at a presulfidation temperature of 873 K.

Fig. 6 presents the W  $L_3$ -edge XANES spectra of 10 and 20% WS<sub>2</sub>/SiO<sub>2</sub> sulfided at 673 and 773 K for examination of the chemical state of W in the sulfided catalysts, since it is well known that supported-W oxides are not easily sulfided compared with the Mo oxide counterparts [5,13,32,33]. The XANES spectra of 10% WO<sub>3</sub>/SiO<sub>2</sub> and a crystalline WS<sub>2</sub> reference are also shown for comparison. WS<sub>2</sub>/SiO<sub>2</sub> (773) showed a XANES spectrum very close to the one due to crystalline WS<sub>2</sub> regardless of the W loading, indi-

cating complete sulfidation of W oxides to WS<sub>2</sub> particles. On the other hand, the white line around 10208 eV for the  $WS_2/SiO_2$  (673) catalysts was much broader than that for crystalline  $WS_2$ . It is assumed that the peak is broadened by a superposition of the peaks due to W oxides and sulfides, judging from the peak position (10210 eV) for WO<sub>3</sub>/SiO<sub>2</sub>. The peak was successfully deconvoluted into two peaks due to WS<sub>2</sub> and W oxides, as exemplified in Fig. 6B for 20%  $WS_2/SiO_2$  (673), allowing the estimation of the sulfidation degree of W. The contribution of the WS<sub>2</sub> peak for 20%  $WS_2/SiO_2$  (673) was 56% and much lower than that for 10%  $WS_2/SiO_2$  (673) (81%). Since the sulfidation degree of W was lower at a higher W content, it is assumed that large  $WO_3$  crystallites are only partially sulfided to  $WS_2$  particles. The XPS spectrum of the W4 f level for 20%  $WS_2/SiO_2$ (673) (not shown for brevity) showed a clear contribution of W oxides (W4 f7/2: 36.3 eV) to WS2 (32.6 eV). A deconvolution of the spectrum indicated that 21% of W remained unsulfided after the presulfidation at 673 K, suggesting the formation of the surface WS<sub>2</sub> layers around W oxide cores. Similar phenomena were observed by XPS for the sulfidation of poorly dispersed MoO<sub>3</sub>/SiO<sub>2</sub> [34]. On the basis of extensive physicochemical characterizations, van der Vlies et al. [35,36] have proposed a detailed mechanism of sulfidation of crystalline monoclinic WO<sub>3</sub> and WO<sub>3</sub>  $\cdot$  H<sub>2</sub>O to WS<sub>2</sub>, involving the formations of hydrogen tungsten bronze, shear oxides, and various oxysulfide intermediates, and eventual transformations of the oxysulfidic phases to WS<sub>2</sub>.

We measured the magnetic property of the Co species in the CVD-Co/Mo(W)S<sub>2</sub> catalysts to study the effects of the support and metal sulfide component on the structure and electronic state of Co [18,19]. Fig. 7 presents the effective magnetic moment and magnetic susceptibility  $\chi$  of Co for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (773) as a function of temperature. The catalyst was treated in a H<sub>2</sub> stream at 673 K for 30 min after the second sulfidation. With decreasing temperature, the magnetic susceptibility  $\chi$  of Co increased and had a maximum around 15 K, followed by a sharp decrease at a lower

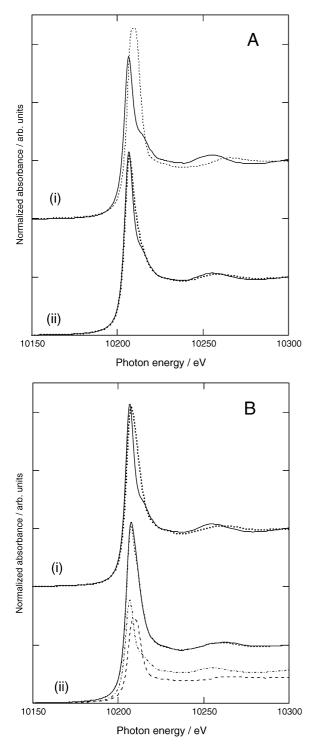


Fig. 6. W  $L_3$ -edge XANES spectra for 10% WS<sub>2</sub>/SiO<sub>2</sub>, 20% WS<sub>2</sub>/SiO<sub>2</sub>, and references. (A) (i) polycrystalline WS<sub>2</sub> (solid line) and 10% WO<sub>3</sub>/SiO<sub>2</sub> (dotted line), (ii) 10% WS<sub>2</sub>/SiO<sub>2</sub> (673) (dotted line) and 10% WS<sub>2</sub>/SiO<sub>2</sub> (773) (solid line). (B) (i) 20% WS<sub>2</sub>/SiO<sub>2</sub> (673) (dotted line) and 20% WS<sub>2</sub>/SiO<sub>2</sub> (773) (solid line), (ii) deconvolution of the XANES spectrum of 20% WS<sub>2</sub>/SiO<sub>2</sub> (673) (solid line, observed; dotted line, convoluted spectrum) into two components (WS<sub>2</sub>, dash-dot line; WO<sub>3</sub>, dash line).

temperature. This temperature dependency of  $\chi$  is a typical antiferromagnetic behavior. A similar magnetic property of Co was previously reported for CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673)

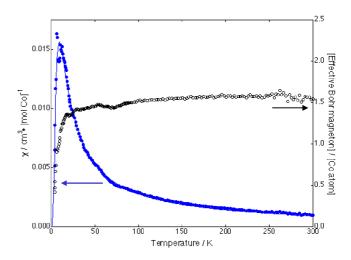


Fig. 7. Magnetic susceptibility  $\chi$  and effective magnetic moment ( $\mu_B$ ) of Co as a function of temperature for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (773) treated in an H<sub>2</sub>-stream at 673 K. The best fitting curve, assuming a dinuclear Co sulfide cluster, for the observed magnetic susceptibility is also shown.

Table 2

Magnetic parameters<sup>a</sup> obtained by fitting the experimental magnetic susceptibility of Co assuming the formation of dinuclear Co clusters on the edge of  $MoS_2$  or  $WS_2$  particles

Catalyst	Mo(W)O <sub>3</sub> content (wt%)	Presulfidation temperature (K)	Ј (К)	α
CVD-Co/MoS <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> <sup>b</sup>	13	673	-7.1	0.84
CVD-Co/MoS <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> <sup>b,c</sup>	13	673	-6.9	0.96
CVD-Co/WS <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> <sup>c</sup>	22	773	-7.4	0.77
CVD-Co/MoS <sub>2</sub> /SiO <sub>2</sub>	10	673	-8.5	1.2
CVD-Co/WS2/SiO2c	20	673	-9.6	0.98

<sup>a</sup> Eq. (1).

<sup>b</sup> Previous results [19].

<sup>c</sup> H<sub>2</sub>-trated at 673 K for 30 min.

[18,19]. With CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub>, the magnetic susceptibility  $\chi$  of Co was reasonably fitted by Eq. (1) [37], assuming the formation of a dinuclear Co sulfide cluster on the edge of MoS<sub>2</sub> particles (spin-pair model)

$$\chi = \alpha N_{\rm A} g^2 \mu_{\rm B}^2 / k_{\rm B} T \left[ 3 + \exp(-2J/k_{\rm B}T) \right], \tag{1}$$

where  $N_A$  is Avogadro's constant,  $\mu_B$  is the Bohr magneton, g is the gyromagnetic factor (assumed to be 2 here),  $k_B$  is Boltzmann's constant, J is the magnetic interaction strength (defined by  $H = -2JS_1 \times S_2$ ), and  $\alpha$  is the fraction of the paramagnetic spin per Co atom. It was found that the temperature dependency of the observed  $\chi$  for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (773) was also rationally fitted by using the theoretical Eq. (1) as shown in Fig. 7. The fitting parameters, J and  $\alpha$ , are summarized in Table 2 together with the previous results for CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> [19]. The effective magnetic moment for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> is close to the theoretical value for Co(II) at a low spin state,  $1.73\mu_B$ , at around 300 K. As presented in Table 2, the magnetic property of CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> was not changed by the treatment in a H<sub>2</sub> stream at 673 K for 30 min [19]. It should be noted that the *J* values for CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673) and CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673) are the same within experimental accuracy ( $\pm 0.5$  K).

With CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (673) and CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (673), it was found that the temperature dependency of the magnetic susceptibility  $\chi$  of Co was also reasonably fitted by Eq. (1) (not shown for brevity). The magnetic parameters are presented in Table 2. It seems that the *J* values for the SiO<sub>2</sub>-supported Mo and W catalysts are rather close to each other but are slightly greater in the absolute value than those for the Al<sub>2</sub>O<sub>3</sub>-supported catalysts. The values of  $\alpha$  are close to unity for the CVD catalysts regardless of the support and catalyst composition, indicating that the dinuclear Co species are preferentially formed in these catalysts by the CVD technique. The effective magnetic moments around 300 K indicate that Co(II) species in the CVD catalysts are in a low spin state characteristic of sulfur environments even after the H<sub>2</sub> treatment at 673 K.

#### 4. Discussion

## 4.1. Selective formation of the Co–W–S phase and its structure

In our previous study on various CVD-Co/MoS<sub>2</sub>/support (support: Al<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub>, ZrO<sub>2</sub>, or SiO<sub>2</sub>) catalysts, it has been shown by XPS [27] and Co-K-edge XANES [30] spectra that the Co sulfide species, admitted by the CVD technique with  $Co(CO)_3NO$ , preferentially interact with  $MoS_2$  particles. Furthermore, the amount of anchored Co as expressed by the Co/Mo mole ratio was proportional to the amount of NO adsorption (NO/Mo mole ratio) on the MoS<sub>2</sub>/support catalyst, substantiating the conclusion that the Co sulfide species are located on the edge of  $MoS_2$  particles [27–29], since it is well established that NO molecules adsorb selectively on the edge of MoS<sub>2</sub> particles. These results allowed us to conclude that the Co-Mo-S phase [5] or the Co-Mo-S structure, in which Co atoms interact with a MoS<sub>2</sub> edge, is preferentially and fully formed by the CVD technique. The FTIR spectra of NO adsorption on CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> substantiated the conclusions [29].

Fig. 4 shows a proportional correlation between the Co/W and NO/W ratios for the WS<sub>2</sub> catalysts regardless of the W loading, sulfidation temperature, and support, demonstrating that the Co–W–S phase is selectively formed by the CVD technique as observed for the Co–Mo–S phase. The selective formation of Co–WS<sub>2</sub> interaction species was suggested by the Co-*K*-edge XANES spectra for CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> [25]. The value of  $\alpha$  in Table 2 shows the fraction of paramagnetic spin per Co atom accommodated in the CVD-Co/Mo(W)S<sub>2</sub> catalysts. As discussed elsewhere [18,19], the Co species in the Co–Mo–S phase are described as a dinuclear Co sulfide cluster model with an antiferromagnetic property, since Co<sub>9</sub>S<sub>8</sub> shows a Pauli paramagnetic behavior ( $\chi = 1.2 \times 10^{-4}$  cm<sup>3</sup> mol<sup>-1</sup> Co<sup>-1</sup>) [38]. Accord-

ingly, the selective formation of the Co–W–S phase is confirmed by the finding that the value of  $\alpha$  is essentially unity within experimental accuracy.

As shown in Fig. 7, the magnetic property of CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (773) can be described by Eq. (1), indicating that the Co species in the Co–W–S phase form a dinuclear Co sulfide structure, as observed for the Co–Mo–S phase in CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673) [18,19]. With the SiO<sub>2</sub>-supported CVD-Co/Mo(W)S<sub>2</sub> catalysts, the same conclusions can be derived from Table 2. Together with the spectroscopic and adsorption results for CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (Fig. 4), the value of  $\alpha$  in Table 2 and Co-*K*-edge XANES [25] strongly indicate that the antiferromagnetically coupled Co atoms in CVD-Co/WS<sub>2</sub> catalysts are in sulfur environments and are situated at the edge of WS<sub>2</sub> particles.

The parameter J, which shows the magnitude of the magnetic interactions between the adjacent Co atoms of CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (773), is very close to that of CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673), suggesting that the local structure and electronic state of the Co(II) species in the Co-W-S phase and the Co–Mo–S phase are very similar to each other. The value of J for CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> is slightly larger than or rather close to that for CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> within the experimental accuracy ( $\pm 0.5$  K). On the other hand, the values of J for the SiO<sub>2</sub>-supported catalysts are slightly but significantly greater in the absolute value than those for the Al<sub>2</sub>O<sub>3</sub>-supported counterparts. Although it is assumed that the difference in the J value resulted from the differences in the local structure and electronic state of the Co species in the Co-Mo(W)-S phase, caused by the interactions with the support, we have to await more systematic experiments and theoretical calculations for a more detailed discussion on the Co-Mo(W)-S structure, magnetic property, the effect of support, the effect of pretreatment (evacuation at 673 K in the present study), and the TOF of the HDS reaction. However, it is worth noting here that the magnetic property of the Co-Mo(W)-S phase can differentiate slight changes in the nature of the Co species between the SiO<sub>2</sub>- and Al<sub>2</sub>O<sub>3</sub>supported catalysts, in contrast to XPS (the same  $Co2p_{3/2}$ binding energy [27]).

### 4.2. Intrinsic activity of the Co–Mo–S phase and the Co–W–S phase

As shown in Fig. 5 and Table 1, the TOF values of the thiophene HDS on the Co–Mo–S phase and the Co–W–S phase strongly depend on the support, metal sulfide component, and presulfidation temperature. With the CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> and CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> catalysts sulfided at 673 K, the TOF values on the Co–Mo–S phase and the Co–W–S phase are identical under the present reaction conditions. Louwers and Prins [26] suggested that the intrinsic activities of the Ni–Mo–S phase and the Ni–W–S phase supported on activated carbon are identical, on the basis of the activity increase with the addition of Ni. The present results directly demonstrate, for the first time, that

the TOF values on the Co–Mo–S phase and the Co–W–S phase supported on Al<sub>2</sub>O<sub>3</sub> are identical, on the basis of the actual number of the active sites for the HDS of thiophene. Topsøe et al. defined two types of the Co–Mo–S phase depending on the intrinsic activity, Co–Mo–S Type I and II [5,39,40]. Co–Mo–S Type II, which is formed by high-temperature sulfidation (875–1275 K), is about two times more active for the HDS of thiophene than Co–Mo–S Type I formed by low temperature sulfidation (675 K). Since CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> and CVD-Co/WS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> were presulfided at 673 K, it is concluded that Co–Mo–S Type I and Co–W–S Type I show the same TOF of the HDS of thiophene, in conformity with a close *J* value in Table 2.

The TOF over CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (673) is 1.8 times that of CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673), in agreement with the previous study [27]. On the basis of the results, we previously concluded that there was formation of Co-Mo-S Type II in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (673). However, as shown in Fig. 5 and Table 1, the TOF over CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> is increased by the presulfidation at 773 K and further increased, although to a lesser extent, by the sulfidation at 873 K. The TOF over CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (873) is 2.3 times that over CVD-Co/MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> (673); the extent of the increase in TOF is consistent with that reported by Topsøe et al. [5,39] for the shift from Co–Mo–S Type I to II. Accordingly, it is concluded that the Co-Mo-S phase in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (873) is classified as "real" Co-Mo-S Type II defined by Topsøe et al. [5,37]. The Co-Mo-S phase in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (673), with a TOF intermediate between those of Co-Mo-S Type II and I, is designated as Co-Mo-S pseudo-Type II hereinafter for convenience. In the present study, the TOF calculated here is averaged over the Co-Mo-S structures existing in the catalysts. However, it is unlikely that Co-Mo-S pseudo-Type II in CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (673) is a simple mixture of Co-Mo-S Type I and II, since the TOF on Co-Mo-S pseudo-Type II is independent of the Mo content or MoS<sub>2</sub> particle size within the experimental accuracy and since MoS<sub>2</sub> particles only weakly interact with SiO<sub>2</sub> surface [41,42] and are completely sulfided at 673 K [34,43,44].

The origin of Co-Mo-S Type I has been attributed to strong interactions between Mo oxides and support and, accordingly, to incomplete sulfidation of Mo oxides to MoS<sub>2</sub> particles and their resultant single-slab structure with residual anchoring bonds with the support such as Mo-O-Al, whereas that of Type II has been attributed to complete sulfidation and the formation of stacked MoS<sub>2</sub> particles [5,39]. The formation of MoS<sub>2</sub> stackings (averaged stacking number 2.1) was confirmed by TEM for 10%  $MoS_2/SiO_2$  (673). It is therefore assumed that full sulfidation and the formation of stacked MoS<sub>2</sub> particles are not enough to form Co-Mo-S Type II. On the basis of the dependence of the TOF on the support and presulfiding temperature (Fig. 5 and Table 1), we instead propose that Co-Mo-S pseudo-Type II and Type II are formed on the edge of fully sulfided MoS<sub>2</sub> particles with no strong interactions with the support, in contrast to Co–Mo(W)–S Type I, and that pseudo-Type II is correlated with a distorted structure of  $MoS_2$  particles, whereas Type II is correlated with a well-crystallized  $MoS_2$  structure. It seems rational to assume that the crystallinity of  $MoS_2$ particles is mainly determined by the sulfidation temperature at a fixed  $H_2S/H_2$  pressure. Recently, Kooyman et al. [45] clearly showed by HRTEM that the crystallinity of supported  $MoS_2$  particles is improved by sulfidation at 873 K compared with those formed at 673 K.

With the W-based catalysts, we propose the formation of Co-W-S pseudo-Type II in CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (773), in which  $WO_3$  is completely sulfided to  $WS_2$ , as indicated by the W  $L_3$ -edge XANES spectra in Fig. 6. The TOF on CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> is significantly increased by the presulfidation of WS<sub>2</sub>/SiO<sub>2</sub> at 873 K, suggesting the formation of Co-W-S Type II simply on the basis of a 2.5 times increase in TOF relative to Co-W-S Type I observed for CVD- $Co/WS_2/Al_2O_3$  (673). It is confirmed that full sulfidation (Fig. 6) and the formation of WS<sub>2</sub> stackings (averaged stacking number 3.7) are not enough to form Co-W-S Type II. The formation of Co–W–S pseudo-Type II requires a higher presulfidation temperature (773 K) than Co-Mo-S pseudo-Type II (673 K). This is correlated with a higher difficulty of the full sulfidation of W oxides than that of Mo oxides to the corresponding sulfides, as evidenced by the XANES spectra in Fig. 6. With CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (673), the relatively low TOF may have resulted from the incomplete sulfidation of W oxides (XANES in Fig. 6 and XPS). It should be noted that there remains a possibility that a presulfidation temperature higher than 873 K is needed for the formation of "real" Co-W-S Type II, in contrast to the formation of Co-Mo-S Type II at 873 K [10,11].

As for the Al<sub>2</sub>O<sub>3</sub>-supported catalysts, it is assumed by definition [5,39] that Co–Mo–S Type I and Co–W–S Type I are formed, when presulfided at 673 K. These phases show identical TOFs. When  $Mo(W)S_2/Al_2O_3$  is presulfided at 773 K, the TOF is increased by 10–20%; this is ascribed to partial formation of Co–Mo(W)–S pseudo-Type II in these catalysts.

The TOF values increase in the order Co–Mo(W)–S Type I (7.1–7.3 h<sup>-1</sup>) < Co–Mo(W)–S pseudo-Type II (11.2– 12.8 h<sup>-1</sup>) < Co–Mo(W)–S Type II (16.5–17.9 h<sup>-1</sup>). The different intrinsic activities of these phases stem from a slight difference in the local structures and electronic states of the Co species in the active phase. If we compare the TOFs in Table 1 and the *J* values in Table 2, it seems that these values are closely correlated with each other. However, a further systematic study is needed to establish the correlation between the magnetic property and TOF.

It is well known that only a weak synergetic effect appears between Co and W sulfides compared with the other combinations [23]. Kishan et al. [24] and Kubota et al. [25] have demonstrated that the HDS activity of Co–W catalysts is greatly increased by the addition of a chelating agent to impregnation solutions containing Co, suggesting that the weak catalytic synergy between Co and W sulfides is caused

by a difference in the sulfidation temperatures between Co and W precursors that is greater than those between Co and Mo counterparts. The present results clearly demonstrate that if Co–W sulfide catalysts were prepared properly, strong synergetic effects would be generated, even between Co and W sulfides, as observed for Co–Mo sulfide catalysts. The fact in Fig. 1 that the HDS activity of CVD-Co/WS<sub>2</sub>/SiO<sub>2</sub> (873) is lower than that of CVD-Co/MoS<sub>2</sub>/SiO<sub>2</sub> (873) does not result from a lower intrinsic activity of the Co–W–S phase than that of the Co–Mo–S phase, but is due simply to a lower edge dispersion of WS<sub>2</sub> particles compared with that of MoS<sub>2</sub> particles, as suggested by the NO/Mo(W) ratio in Fig. 3 and the TEM observations.

Close similarities in TOF between the Co–Mo–S phase and the Co–W–S phase may imply the reaction mechanism of HDS on the Co-Mo(W)-S phase or the genesis of the catalytic synergy between Co and Mo or W sulfides. There are mainly two proposed reaction mechanisms of HDS on the Co-Mo(W)-S phase; one model assumes that the reaction takes place on coordinatively unsaturated Co sites and that  $MoS_2(WS_2)$  edges provide only a "support" to form highly active Co species (Co-only model) [26,46,47], and another model assumes that the sulfur vacancy between Co and Mo(W) atoms, Co– $\Box$ –Mo(W), provides an active site and the Co and Mo(W) atoms simultaneously participate in the activation of organic sulfur compounds, thus generating catalytic synergy [45]. As discussed by Louwers and Prins [26], the present results suggest the reaction mechanism in which coordinatively unsaturated Co atoms play a major role under the present reaction conditions (623 K). If the reactants were activated on the active sites, Co-D-Mo(W), the intrinsic activities of the Co-Mo-S phase and the Co-W-S phase would be expected to be significantly different, reflecting the difference in the bond energy between Mo-S and W-S bonds, the latter being greater than the former [48]. However, the involvement of the  $MoS_2$  or  $WS_2$ edge or corner sites, in part, in the reaction (e.g., through the activation of hydrogen and formations of Mo(W)-SH groups [49]) cannot be ruled out.

#### 5. Conclusions

In the present study, we tried to compare the intrinsic catalytic activities of the Co–Mo–S phase and the Co–W–S phase for the HDS of thiophene. The CVD technique was used to prepare supported Co–Mo and Co–W sulfide catalysts, in which all of the Co atoms form active sites. The catalysts were characterized by NO adsorption, TEM, XPS, W  $L_3$ -edge XANES, and the measurements of the magnetic susceptibility and effective magnetic moment of Co. The salient findings of the present study are as follows:

 The Co–W–S phase supported on SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> was selectively formed by the CVD technique with the use of Co(CO)<sub>3</sub>NO, as shown previously for the Co–Mo–S phase. The Co–W–S phase showed an antiferromagnetic property.

- 2. The TOF of the Co–Mo–S phase and the Co–W–S phase increased as the presulfidation temperature of MoS<sub>2</sub>/SiO<sub>2</sub> and WS<sub>2</sub>/SiO<sub>2</sub> increased from 673 to 873 K, whereas it was independent of the loading of Mo and W (5–20 wt% MoO<sub>3</sub> and WO<sub>3</sub>).
- 3. The TOFs on the Co–Mo–S phase and the Co–W–S phase were very close for the HDS of thiophene; that is, Co–Mo–S Type I ≈ Co–W–S Type I < Co–Mo–S Type II ≈ Co–W–S Type II. The magnetic results suggest that the TOF is changed by a slight modification in the local structure and electronic state of Co in the Co–Mo(W)–S phase.</p>
- It is suggested that Co–Mo(W)–S Type II is not necessarily formed by full sulfidation and stackings of Mo(W)S<sub>2</sub> slabs, but by a high-temperature treatment.

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